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An empirical potential for silicon under conditions of strong electronic excitation

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We present an empirical potential developed for silicon under conditions of strong electronic excitation. We show the essentially athermal nature of the melting transition when the electronic temperature is extremely high. The resulting liquid is shown to be distinct from ordinary liquid silicon. For less intense excitations, we determine the thermal melting temperature and demonstrate the possible existence of a regime where ordinary thermodynamic melting can occur but at a reduced temperature T_m . We show laser-induced softening of the lattice can lead to lattice cooling for very short time scales (~100 fs), an effect never before recognized. © 2010 American Institute of Physics. [doi:10.1063/1.3499296]

For some time now there has been a great deal of attention given to the processing of materials using laser pulses. Recent advances have enabled the generation of ultrashort (100 fs or less) laser pulses that can generate local carrier densities high enough ($\sim 10^{22}$ cm⁻³) to cause a destabilization of the lattice. The rapid production of liquid by athermal transitions has been directly observed by several groups, and in situ x-ray diffraction has provided direct visualization of the atomic motion in indium antimonide.¹ While the experimental evidence for the role of athermal processes is quite strong, there are many unanswered fundamental questions relating to the details of the structural evolution and the relevant time scales. For example, one interesting possibility is that a regime exists where thermal melting transitions occur at a temperature lower than the ordinary melting transition due to softening of the lattice. It might therefore be possible to observe a gradual transition from thermal transitions to purely athermal transitions.

At present, the available theoretical tools are unable to completely resolve these questions. Approaches based on finite-temperature density-functional theory (DFT) (Refs. 2 and 3) and tight-binding⁴ have shown that electronic excitation in silicon leads to melting and a resultant liquid with properties quite distinct from ordinary liquid silicon. Recently it has been shown that electronic excitation in silicon can lead to lattice instabilities as exhibited by transverse acoustic (TA) phonon modes with imaginary frequencies.⁵ However, DFT approaches are limited in their ability to directly model annealing, melting, or ablation. Instead, molecular-dynamics simulation with empirical potentials is often applied to study laser processing and ablation.⁶ However, empirical potentials have not been developed to capture the effect of lattice instabilities due to electronic excitation.

In this letter, we present an empirical potential used to model silicon when the electrons are in a highly excited state. This same basic approach has been applied to metals.⁷ The parameters of the potential are determined by fitting to finite-temperature DFT calculations. This potential is then used to explore the dynamics of silicon after the excitation to very high electronic temperatures T_e .

To provide a database for the empirical potential, we performed finite-temperature DFT calculations using the ABINIT code.⁸ The conditions for our calculations were quite close to that given in Ref. 5. In particular, the local-density approximation was used, with the 3s, 3p, and 3d orbitals included for the valence electrons. The energy cutoff for the plane-wave expansion was taken to be 544 eV, and an 8×8×8 Monkhorst-Pack mesh of k-points was used. Result were obtained for diamond, fcc, bcc, and simple-cubic lattices at seven different volumes each. Calculations were performed for electronic temperatures between $k_B T_e = 0$ eV and $k_BT_e=2.50$ eV. Starting at $k_BT_e=0.2$ eV, the calculations were performed in increments of 0.05 eV, thereby generating results for 48 different temperatures. We obtain in each case the cohesive free energy. For the diamond lattice, we obtained results that are in good agreement with prior DFT calculations. 2,3,5

The empirical potential is based on a recently published modified Tersoff formalism (called the MOD potential here and in Ref. 9), which has been shown to provide an excellent description of the phase behavior and kinetics.^{9,10} For k_BT_e =0, we use the original formulation except with larger cutoffs (R_1 =3.1 Å and R_2 =3.4 Å) and α =1.90 chosen to give a melting temperature T_m =1688 K close to experiment. For

TABLE I. Parameters used for the empirical potential for different values of T_e . The notation follows Ref. 9. Parameters not given in the table are unchanged from the original potential. The constant term F_0 is the free-energy of an isolated silicon atom at finite T_e . We also include the phonon frequency at the X-point (in cm⁻¹) with the lattice parameter a=5.429 Å and the DFT value in parenthesis for comparison.

$\overline{k_B T_e \text{ (eV)}}$	0.50	1.30	2.15	2.50
A (eV)	2 944.068 5	2 179.984 4	1 827.473 1	1 761.681 2
<i>B</i> (eV)	92.323 2	53.836 9	35.786 8	31.868 6
λ_2 (Å ⁻¹)	1.223 828	0.769 646	0.370 933	0.243 576
$\eta imes \delta$	0.610 977	0.785 708	0.867 687	0.906 381
α	1.679 913	1.328 494 3	1.081 435	1.027 233
c_1	0.214 947	0.637 945	1.068 782	1.155 003
c_2	533 198.46	172 192.94	164 461.57	160 964.28
c_4	1.090	1.234	1.387	1.450
F_0 (eV)	0.000 2	0.137 48	0.892 8	1.340 0
$\omega_{\mathrm{TA}}(X)$	235 (135)	63 (69)	-101 (-97)	-124 (-108)

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FIG. 1. (Color online) Cohesive free energy per Si atom determined by finite-temperature DFT and the fit empirical potential at two values of T_e . The DFT results are indicated by symbols (diamonds: diamond; triangles: simple-cubic; squares: bcc; and circles: fcc), and the empirical potential is shown by lines (solid line: diamond; dotted line: simple-cubic; dashed line: bcc; and broken line: fcc).

finite T_e , the parameters were fit to reproduce the DFT cohesive energies for the fcc, bcc, simple cubic, and diamond structures at several different volumes. Some parameters were found to not be critical, and hence were held fixed. Because the potential for $k_B T_e = 0$ was not fit to the DFT, we fit instead to changes in the free energy curves as a function of electronic temperature T_e . Also, the DFT energies were shifted to give the experimental value $E_{coh}=4.63$ eV in the equilibrium structure. In Table I, we present the parameters used in the calculations described here. We include in Table I, a constant F_0 that is added to the empirical potential to take into account the fact that the free energy of an isolated silicon atom will also depend on T_e . In Fig. 1 we show a comparison of the cohesive free energies predicted by DFT and the empirical potential.

We show the dependence of the $T_{ion}=0$ K bulk modulus (Fig. 2) and lattice parameter (Fig. 3) on the electronic temperature T_e . To obtain the lattice parameter and bulk modulus from the DFT calculations, we fit the data to the Birch-Murnaghan equation of state. Another important characteristic is the dependence of the phonon frequencies on T_e . It has previously been shown that the TA branch in diamond silicon is unstable at $k_B T_e = 2.15$ eV.⁵ We give in Table I the TA phonon frequencies ω_{TA} at the X-point with a comparison to the DFT predictions. In each case, the lattice parameter is fixed at the value a=5.429 Å. While the values of ω_{TA} are significantly larger than the DFT results for low T_e , the over-



FIG. 3. (Color online) Lattice parameter of the diamond structure computed for the empirical potential (red squares) and DFT (black circles).

all qualitative agreement with DFT is quite good and in fact improves for higher T_e .

To simulate the effect of an intense laser pulse, we first equilibrate an $8 \times 8 \times 8$ cell with 4096 Si atoms at T_{ion} = 300 K with the empirical potential for $k_B T_e = 0$ for 11.05 ps. After this step, we instantaneously change the empirical potential to correspond to strong excitation and continue the simulation at constant energy and volume. We chose constant volume boundary conditions to compare with existing DFT results with the same boundary conditions.^{2,3} As a result of the excitation, the temperature evolves away from T_{ion} = 300 K. In Fig. 4 we show the evolution of T_{ion} with time after the excitation. We find that for $k_B T_e = 2.15$ eV and $k_B T_e = 2.50$ eV, the lattice melts and T_{ion} strongly increases within about ~ 100 fs. We find final temperature T_{ion} ≈ 580 K when $k_B T_e = 2.15$ eV, significantly lower than $T_{\rm ion} \approx 1700$ K found in previous DFT calculations.^{2,3} We believe that the difference with the DFT results is due to the fact that the starting MOD potential over predicts the energy difference between diamond and the highly coordinated fcc and bcc lattices. Since the fits at finite T_e are done to reproduce the changes with T_e , we have a similar error at all values of T_e . Hence we predict a smaller change in energy (by less than 0.1 eV per atom) when the system transitions to the highly coordinated liquid, and hence the increase in T_{ion} is somewhat smaller. However, we find that at $k_B T_e$ =2.50 eV that $T_{\rm ion} \approx 950$ K. Interestingly, we find that for $k_B T_e = 1.30$ eV, the system does not melt and in fact T_{ion} slightly decreases to $T_{\rm ion} \approx 230$ K. At $k_B T_e = 1.30$ eV, there is no lattice instability but instead a significant softening of



FIG. 2. (Color online) Bulk modulus of the diamond structure computed for the empirical potential (red squares) and DFT (black circles).



FIG. 4. (Color online) Ionic temperature T_{ion} plotted as a function of time after electronic excitation to $k_BT_e=1.30$ eV, $k_BT_e=2.15$ eV, and $k_BT_e=2.50$ eV.



FIG. 5. (Color online) Radial distribution function for melted silicon after excitation at $k_B T_e$ =2.15 eV (solid red line). For comparison we show ordinary liquid Si ($k_B T_e$ =0) at T=1700 K (dotted black line).

all phonon modes. Hence, the atomic displacements become larger but the average kinetic energy decreases due to weaker restoring forces. However, the excited system is under fairly high stress and we expect the temperature to rise somewhat in a constant pressure simulation.

For the liquid resulting from the excitation, we find a distinct structure compared to ordinary liquid silicon. In Fig. 5 we show the radial distribution function g(r) for the k_BT_e = 2.15 eV simulation in comparison to a simulation with k_BT_e =0 and T_{ion} =1700 K. The first peak in g(r) occurs at 2.55 Å in the excited silicon, compared to 2.42 Å in the ordinary liquid. In contrast to DFT predictions,^{2,3} we see a sharp minimum at 3.13 Å. Using this minimum as a cutoff, we find a coordination number of 5.31. However, if we increase the temperature to T_{ion} =1700 K, Fig. 6 shows that this minimum largely goes away. In this case, using 4.0 Å as a cutoff, the coordination number is 12.71, which compares quite well with the DFT value in the range 11–13.^{2,3}

Finally we obtain the melting temperature T_m =1278 K for the $k_B T_e$ =0.5 eV case. To determine T_m , we simulate coexistence of liquid and crystal.¹⁰ This demonstrates that electronic excitation might lead to a thermal transition at a strongly modified temperature.

In summary, we have developed an empirical potential for silicon that depends on the electronic temperature T_e . We



FIG. 6. (Color online) Radial distribution function for $k_B T_e = 2.15$ eV (solid red line) and $k_B T_e = 0$ (black dotted line) both computed at $T_{ion} = 1700$ K.

have elucidated some aspects of melting in conditions of strong laser excitation. For relatively weak excitation, we find that cooling can occur for short time scales.

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