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Multisite nature and efficient lasing at 1041 and 1302 nm in Nd³⁺ doped potassium yttrium fluoride

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Lasing of Nd³⁺ ions in different crystallographic sites of KYF₄ has been demonstrated to occur at different wavelengths. Low threshold, high efficiency lasing performance at 1041 and 1302 nm were achieved under selective pumping conditions. The stimulated emission cross section at 1041.2 nm was estimated to be 2.6×10^{-19} cm² and the radiative lifetime was found to be 600 μ s. © 1995 American Institute of Physics.

KYF₄ (KYF) was considered as an attractive Nd³⁺ host for diode laser pumping¹ because Nd³⁺ ions in this crystal have a broader absorption band in the 800 nm region and a longer fluorescence lifetime than in either Y₃Al₅O₁₂ (YAG) or LiYF₄ (YLF). It was also an interesting material because the lasing wavelength of this crystal was among the shortest wavelengths available for Nd³⁺ ions in crystals. However, earlier work² did not show much promise for this crystal. A slope efficiency of only 30% was obtained with a high threshold of 300 mW for cw Ti:sapphire laser pumped laser operation.² In this letter, we report the multiwavelength lasing of Nd³⁺ ions resulting from the multisite nature of KYF₄. By pumping the crystal at selected wavelength, low threshold, high efficiency lasing was achieved at both 1041 and 1302 nm. The discrepancies between the present results and those previously reported will be discussed.

High quality Nd³⁺:KYF₄ crystals were grown by the conventional Czochralski method. The melting point of KYF₄ is slightly higher than 800 °C. The melt composition was 43% YF₃ and 57% KF. The typical crystal growth parameters are pull rate=0.5 mm/h and rotation=15 rpm. The growth conditions are discussed in detail elsewhere.³

The room-temperature emission spectra of Nd³⁺:KYF₄ in the 1 μ m region are shown in Fig. 1. They were obtained by exciting a sample containing 1.2 at. % Nd³⁺ with a cw Ti:sapphire laser. The spectral width of the laser was ~ 2 cm⁻¹. The emission in this region consists mainly of two peaks centered at 1041 and 1054 nm. The excitation wavelengths, 799.4 and 804.5 nm, so chosen to maximize the relative intensity of the peak at 1041 nm for curve a and the peak at 1054 nm for curve b in Fig. 1, respectively, correspond to the two main absorption peaks in the 800 nm region as shown in Fig. 2. As seen in Fig. 1 the relative intensity of the two emission peaks changes with excitation wavelength. This indicates that the emissions originate from different electronic centers. The emission dynamics of these two features were studied in a dilute sample containing 0.18 at. % Nd³⁺ at 15 K using a Q-switched Cr:LiSAF laser tuned to 799.6 and 805 nm for 1041 and 1054 nm emission, respectively. As shown in Fig. 3, the emission dynamics differ for

each feature, confirming again that they are of different electronic origin. Our detailed low temperature, selective excitation experiments indicate that each of these two peaks actually consists of a group of lines: two for the 1041 nm peak and four for the 1054 nm peak. However, no difference in the emission dynamics was detected within our experimental accuracy for different lines within each group indicating that lines within each group have similar spectroscopic properties. We therefore classify the groups into two classes according to the previous notation.⁴ Class I for emission in the group peaked at 1041 nm and class II for emission peaked at 1054 nm. The corresponding classification in the absorption spectrum is indicated in Fig. 2. The emission decays shown in Fig. 3 are nearly exponential with a lifetime of 600 and 780 μ s for class I and class II emission, respectively. Careful measurements indicate that the lifetime of each class does not change with temperature. Since these decay times are obtained with a dilute sample at low temperature and they do not change with temperature, they can be regarded as the radiative lifetime of each class. Both selective excitation

FIG. 1. Room-temperature emission spectra of Nd³⁺:KYF₄ in the 1 μ m region using excitation wavelengths of (a) 799.4 nm and (b) 804.2 nm.

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FIG. 2. Room-temperature absorption spectrum of $\text{Nd}^{3+}:\text{KYF}_4$ in the 800 nm region. Arrows indicate the excitation wavelengths used in obtaining the emission spectra shown in Fig. 1.

spectroscopy and dynamics measurements indicated that energy transfer occurs only from class I to class II at temperatures lower than 30 K and occurs in either direction at higher temperatures.⁵

Lasing performance was tested with a 3-mm-long KYF_4 crystal containing 1.2 at.% Nd^{3+} . The room-temperature decay time of this sample was found to be 444 μs for class I and 538 μs for class II emissions. Its end surfaces contained the crystalline *c*-axis and were polished flat and parallel to each other. The crystal was antireflection coated at around 1.05 μm . The laser cavity consisted of a 5 cm radius of curvature high reflector and a flat output coupler with transmission up to 5%. The cavity length was about 4.7 cm. Pulsed and cw operation were studied with excitation by a long-pulse ($\sim 100 \mu\text{s}$) Cr:LiSAF laser and a cw Ti:sapphire laser, respectively. Both pump lasers were tunable with a spectral width of ~ 0.1 nm for the Ti:sapphire laser and ~ 1 nm for the Cr:LiSAF laser. The pump sources were focused into the crystal with a 10 cm focal length lens.

The laser output was π -polarized and occurred at a wavelength which depended on the pump wavelength. Maximum slope efficiencies and lowest thresholds were obtained when pumping at 799.6 nm, which results in lasing at 1041.2

nm. Under this pumping condition and for a 5% output coupler the slope efficiency and threshold were 52% and 9.4 mW for cw operation and 63% and 38 μJ for long-pulse operation. When the pump source was tuned, a small amount from 799.6 nm to either a shorter or longer wavelength, the laser output occurred at 1041.8 nm. When the pump wavelength was tuned further (about 1 nm away) in either direction, lasing output occurred at two wavelengths, 1041.8 and 1054.6 nm, simultaneously. In this case, the lasing threshold was higher and the laser output lower than in the case of single wavelength lasing for the same absorbed power (energy).

The transition of Nd^{3+} between $^4F_{3/2}$ and $^4I_{13/2}$ in KYF_4 results in emission peaked at 1302 nm for class I and at 1307 nm for class II. Our emission measurements indicate that the branching ratio of the $^4F_{3/2} \rightarrow ^4I_{13/2}$ for transition to the $^4F_{3/2} \rightarrow ^4I_{11/2}$ transition is quite high, in contrast to the previous reported results based on Judd-Ofelt calculations.² The peak intensity ratio of the 1.3 μm to the 1 μm emission was measured to be 0.28 for class I and 0.40 for class II. These ratios are quite high compared with the same ratios in most 1.3 μm laser materials. We therefore, tested the 1.3 μm lasing using the same crystal mentioned above and a 5% output coupler. In this case, lasing occurred at 1302 and 1307 nm either separately or simultaneously depending on the pump wavelength. The best performance was achieved at 1302 nm when pumped at 799.6 nm. The long-pulse threshold in this case was 190 μJ and a slope efficiency of higher than 50% was obtained. $\text{Nd}^{3+}:\text{KYF}_4$ performs very well at 1302 nm considering that the quantum-limited slope efficiency in this case is only about 60%. However, cw lasing was not achieved with the maximum pump power of about 200 mW available at the time of this particular experiment.

In order to estimate the stimulated emission cross section of $\text{Nd}^{3+}:\text{KYF}_4$ at 1041.2 nm, comparative lasing studies were made with $\text{Nd}^{3+}:\text{LiYF}_4$ (YLF), a well-known laser crystal. The absorbed power at lasing threshold can be expressed as follows:⁶

$$P_{th} = \frac{\pi h \nu_p \delta}{4 \sigma_e \tau \eta_p} (w_0^2 + w_p^2), \quad (1)$$

where $h \nu_p$ is the pump photon energy, η_p the pump quantum efficiency, τ the upper manifold decay time, σ_e the effective stimulated emission cross section, δ the roundtrip loss which includes both internal and external losses, and w_0 and w_p the beam radii of the laser cavity mode and the pump beam, respectively. The pump quantum efficiency, the number of ions in the upper manifold created by one absorbed photon, should be near unity and therefore about the same for both crystals. In addition, w_0 and w_p are the same for both lasers since the cavity was the same for both lasers. As a result, the effective stimulated emission cross section of $\text{Nd}^{3+}:\text{KYF}$ can be estimated from that of $\text{Nd}^{3+}:\text{YLF}$ by comparing the thresholds, the pump wavelengths, the losses, and the upper manifold decay times of both crystals. Using 5% output coupling a cw laser threshold power of 11.5 mW was obtained for a 1.2% $\text{Nd}^{3+}:\text{YLF}$ crystal lasing at 1.047 μm (pumped at 797 nm) compared to 9.4 mW for $\text{Nd}^{3+}:\text{KYF}$ lasing at 1041.2 nm when pumped at 799.6 nm. The emission decay time of

FIG. 3. Emission decay curves for two different classes of Nd^{3+} ions in KYF_4 at 15 K.

this $\text{Nd}^{3+}:\text{YLF}$ crystal $442\ \mu\text{s}$ is almost the same as that of the $\text{Nd}^{3+}:\text{KYF}$ at $1041.2\ \text{nm}$, $444\ \mu\text{s}$. The double-pass passive loss (δ) can be obtained by studying the measured slope efficiency (η) as a function of output coupler transmission (T) through the following relation⁷

$$\eta = \eta_0 T / (T + \delta), \quad (2)$$

where η_0 is the intrinsic slope efficiency. We found that $\eta_0 = 56.6\%$ and $\delta = 0.5\%$ for $\text{Nd}^{3+}:\text{KYF}$ in cw operation at $1041.2\ \text{nm}$ when selectively pumped at $799.6\ \text{nm}$ and $\eta_0 = 76\%$ and $\delta = 0.45\%$ for the $\text{Nd}^{3+}:\text{YLF}$ crystal. Comparing the above mentioned data for both crystals the effective emission cross section of $\text{Nd}^{3+}:\text{KYF}_4$ at $1041.2\ \text{nm}$ (π polarization) was estimated to be $2.6 \times 10^{-19}\ \text{cm}^2$. We obtained this cross section using the published emission cross section of $\text{Nd}^{3+}:\text{YLF}$, $1.96 \times 10^{-19}\ \text{cm}^2$ at $1.047\ \mu\text{m}$ for π polarization.⁸

It is not surprising that the emission cross section obtained from our selective lasing results is more than 6 times that obtained in Ref. 2 from a Judd–Ofelt calculation using multisite absorption and emission spectra. Since the absorption and emission spectra obtained when the sites are undifferentiated appear to be much broader than those obtained with site selective excitation, the relative peak emission intensity at the lasing wavelength appears to be weaker. As a result, the Judd–Ofelt calculations using such data will result in weaker transition strengths and therefore lower emission cross sections than the intrinsic values for each class. It is also easy to understand our relatively high emission cross section (higher than that of $\text{Nd}:\text{YLF}$) since our site selective spectroscopy indicates that the emission spectrum of each individual site is dominated by a single sharp peak.⁵

Studies of the structure of KYF_4 ^{9,10} revealed that there are six distinct Y^{3+} sites in KYF_4 and that these sites can be classified into two groups, although there is a difference between Refs. 9 and 10 in the classification of the two groups. Our results indicating two classes of dopant sites, are consistent with such results based on structural analysis.

Considering the multisite nature of KYF_4 it is then easy to understand the differences between previous results and ours. Our selective dynamics studies revealed that the two classes of sites have very different radiative lifetimes for the $^4F_{3/2}$ manifold and that these lifetimes do not depend on temperature. The latter is consistent with most Nd^{3+} systems in which the contribution of temperature dependent nonradiative processes to the $^4F_{3/2}$ decay time is negligible. As a result of experiments lacking selective excitation and detection of the individual classes of emitters, Ref. 2 reached the conclusion that the decay time decreases with increasing temperature. Consequently, it implied that the quantum efficiency of Nd^{3+} in KYF_4 is no more than 80% even in the absence of concentration quenching. Such a low value of quantum efficiency would make it difficult to understand our efficient lasing results. Our measured radiative decay time is $600\ \mu\text{s}$ for class I emission and $780\ \mu\text{s}$ for class II emission. Reference 4 gave a decay time of $750\text{--}850\ \mu\text{s}$ for class I and $680\text{--}720\ \mu\text{s}$ for class II. We note that the crystal used in Ref. 4 has a Nd^{3+} concentration of about 0.6 at. % so that the decay time given for class II in Ref. 4 is consistent with our

value for a similarly doped sample. However, the decay time value for class I differs very much from ours. Since a chopped cw Ti:sapphire laser was used in Ref. 4, the sample temperature could easily reach a temperature higher than 30 K even though the nominal experimental temperature was said to be 5 K. When the crystal temperature is higher than 30 K the energy transfer from classes II to I sites occurs as we have observed and mentioned earlier. In this case, the emission dynamics of class I will consist of an initial rise when class II is excited. If a slowly chopped cw source was used in dynamics measurements and if the source was not carefully tuned to excite only class I sites such an initial rise would exist but would go undetected. The result could be a measured decay time that is longer than the real one.

In summary, we have demonstrated the multisite nature of $\text{Nd}^{3+}:\text{KYF}_4$ by spectroscopic and lasing studies. We have found that the two classes of Nd^{3+} ions in KYF_4 have different radiative lifetimes and that these lifetimes do not change with temperature. We have also demonstrated simultaneous lasing from Nd^{3+} ions in different sites and shown that by pumping only one class of dopant ions, low threshold, high efficiency lasing performance can be achieved. The emission cross section for the particular site which lases at $1041.2\ \text{nm}$ was found to be $2.6 \times 10^{-19}\ \text{cm}^2$ and is even higher than that of $\text{Nd}:\text{YLF}$. Our results demonstrate the inapplicability of Judd–Ofelt calculations to multisite systems. We believe that the poor lasing results reported previously² resulted from nonoptimized pumping. The lasing wavelengths, 1041 and 1302 nm, are among the shortest reported for Nd^{3+} ions in crystals, which may enable some wavelength specific applications for this crystal. Also some interesting applications may be found to take advantage of the simultaneous dual-wavelength lasing.

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